

# Effect of off-axis ply orientation on UD fibre microbuckling

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#### Damage Mechanisms under Compression: Fibre microbuckling

Compression failure of laminates occurs by fibre kinking of  $0^{\circ}$  -plies, immediately followed by delamination (catastrophic failure).





Kink band in multidirectional T800/924C laminate



### **0°** fibre microbuckling initiation model



Berbinau, Soutis (1999)



Equilibrium equation (fibre is modelled as a beam on a non-linear foundation)

$$E_{f}I\frac{d^{4}(v-v_{0})}{dx^{4}} + \frac{A_{f}\sigma_{0^{0}-ply}}{V_{f}}\cdot\frac{d^{2}v}{dx^{2}} - 2d_{f}\left\{\left[\frac{d\tau_{zy}}{dy}\right]_{\frac{W}{2}}\right\}\cdot v - A_{f}G\left(\frac{d(v-v_{0})}{dx}\right)\cdot\frac{d^{2}(v-v_{0})}{dx^{2}} = 0$$

Non-linear differential equation that gives the compressive stress  $\sigma_0$  of a 0° -ply in terms of the fibre maximum buckling amplitude V

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#### Free edge effect on fibre microbuckling



Berbinau, Soutis (1999)

The microbuckling model of Berbinau, Soutis accounts for fibre properties, initial fibre waviness, ply interactions (interlaminar stresses), matrix nonlinearity.



- Shear stress components are almost zero around 60°
- Normal component σz is compressive



*Maximum interlaminar edge stresses in a*  $[(\theta /- \theta / \theta_2)_2]_s$  laminate (*strain* =1%)



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#### Fibre wavelength effect on fibre microbuckling triggering

Microbuckling amplitude:

$$\frac{\mathbf{V}}{\mathbf{V}_0} = \mathbf{f}(\mathbf{\sigma}_0)$$



*Microbuckling amplitude V vs.* stress on a 0° -ply  $\sigma_0$  for  $[(30/-30/0_2)_2]_s$ 

Off-axis ply orientation $\theta^{\circ}$	$\sigma_0^*$ Fib $\lambda=10 \cdot d_{fibr}$	re half-waveleng $\lambda = 15 \cdot d_{\text{fibre}}$	gth $\lambda=20 \cdot d_{ ext{fibr}}$
0	968	743	657
30	961	737	637
45	961	723	635
60	975	745	673
75	977	751	680
90	979	753	683

for  $[(\theta/-\theta/\theta_2)_2]_s \sigma_0^*$  depends mostly on:

**G** fibre wavelength matrix non-linearity less on interlaminar shear stresses but may not be the case under fatigue load

![](_page_5_Picture_0.jpeg)

![](_page_5_Picture_1.jpeg)

#### **Compressive strength of a multidirectional laminate**

Stiffness Ratio Method : 
$$\sigma_{\text{lam}} = \frac{\sigma_0^*}{N E_{11}} \sum_{k=1}^{N} n^{(k)} \cdot E_{x0}^{(k)}$$

 $\sigma_0^*$  critical stress for microbuckling initiation in the 0° ply

![](_page_5_Figure_5.jpeg)

Theoretical predictions are conservative

Comparison of experimental and theoretical compressive strength for laminates  $[(\theta/-\theta/0_2)_2]_s$ 

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![](_page_6_Picture_0.jpeg)

## Concluding remarks

![](_page_6_Picture_2.jpeg)

The static compressive failure of UD and MD unnotched <sup>Ce</sup> T800/924C carbon-fibre– epoxy laminates is controlled by fibre microbuckling.

Microbuckling initiates by the elastic bending of the 0°fibres, loaded by resin material in shear.

Its initiation depends on material imperfections, such as resin rich regions, voids and fibre misalignment (waviness).

Model shows the interlaminar shear stress has a small influence on the 0° fibre microbuckling initiation, while the fibre wavelength and matrix non-linearity have a more significant effect.

![](_page_7_Picture_0.jpeg)

![](_page_7_Picture_1.jpeg)

Average measured failure strains  $\varepsilon_{fav}$  for each of the four angles  $\theta$  (30°,45°,60°,75°) examined are scattered between -0.88 and -1%.

Specimens with ±45° surface plies appear to provide the most resistance to fibre microbuckling with  $\varepsilon_{fav} = -0.95\%$ .

The 0°-ply stress in a such laminate is ~1300 MPa compared to 1430 MPa measured for the 100% 0° laminate.

The 10% strength reduction could be due to edge effects and fabrication defects in the form of resin-rich regions and voids introduced at the interface of the axial and off-axis plies.